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### Molecular Crystals and Liquid Crystals

Publication details, including instructions for authors and subscription information:

http://www.tandfonline.com/loi/gmcl20

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Seok Cheol Choi a, Won Suk Yun & Sang Ho Sohn a

<sup>a</sup> Department of Physics , Kyungpook National University , Daegu , 702-701 , Korea

Published online: 08 Jan 2014.

To cite this article: Seok Cheol Choi, Won Suk Yun & Sang Ho Sohn (2013) Enhanced Performance of Dye-Sensitized Solar Cells with Surface-Modified ZnO Nanorods, Molecular Crystals and Liquid Crystals, 586:1, 88-94, DOI: 10.1080/15421406.2013.851536

To link to this article: <a href="http://dx.doi.org/10.1080/15421406.2013.851536">http://dx.doi.org/10.1080/15421406.2013.851536</a>

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Mol. Cryst. Liq. Cryst., Vol. 586: pp. 88–94, 2013 Copyright © Taylor & Francis Group, LLC ISSN: 1542-1406 print/1563-5287 online

DOI: 10.1080/15421406.2013.851536

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## Enhanced Performance of Dye-Sensitized Solar Cells with Surface-Modified ZnO Nanorods

#### SEOK CHEOL CHOI, WON SUK YUN, AND SANG HO SOHN\*

Department of Physics, Kyungpook National University, Daegu 702-701, Korea

The surface destruction of the ZnO nanorod electrodes occurs in the dye-sensitization process, resulting in the extreme decreases of the short-circuit photocurrent density  $(J_{SC})$ , compared to  $TiO_2$  electrodes. To solve these problems, the  $TiO_2$  nanolayer is coated on the ZnO nanorod electrodes using a titanium tetrachloride ( $TiCl_4$ ) assisted sonochemical process, which provides faster surface modification. The  $TiO_2$  nanolayer can form new energy barrier on the ZnO nanorods, which can effectively inhibit the surface charge recombination and improve the energy conversion efficiency of the ZnO nanorod-based dye sensitized solar cells (DSSCs).

**Keywords** Dye-sensitized solar cell; sonochemical; surface modification; TiO<sub>2</sub> nanolayer; ZnO nanorod

#### Introduction

Dye-sensitized solar cells (DSSCs) provide a promising and cost-effective alternative to conventional solar cells because of their high power conversion efficiency, easy fabrication and low cost [1–3]. To date, much work has been performed on DSSCs fabricated with the different nanocrystalline oxide such as  $TiO_2$ , ZnO and  $SnO_2$  to increase their electron transport characteristics. The investigation on ZnO nanorods as an alternative photoelectrode has been intensively carried out due to its band gap and conduction band edge energy which are nearly the same as  $TiO_2$  [2]. Moreover, the electron mobility of ZnO is known to be  $10^7$  times larger than that of  $TiO_2$  [4,5]. However, in the dye-sensitized process, the surface destruction of the ZnO nanorods occurs, resulting in an extreme decrease in photocurrent density ( $J_{SC}$ ) [6]. A decrease in  $J_{SC}$  of ZnO-based DSSCs is attributable to more frequent recombination of photo-generated electrons in the conduction band with oxidized species in electrolytes [7]. In order to solve these problems, it needs the suppression of the recombination occurred in the ZnO electrode/dye and ZnO electrode/electrolyte interfaces.

In this work, we report the  $TiO_2$  nanolayer coated on the ZnO nanorods as a new energy barrier using a  $TiCl_4$  assisted sonochemical process, considering that they can effectively inhibit the recombination of and improve the energy conversion efficiency of the ZnO nanorod-based dye sensitized solar cells (DSSCs).

<sup>\*</sup>Address correspondence to Prof. Sang Ho Sohn, Department of Physics, Kyungpook National University, Sangyuk-dong, Buk-gu, Daegu 702-701, Korea (ROK). Tel.: (+82)53-950-5892; Fax: (+82)53-950-6893. E-mail: shsohn@knu.ac.kr

#### Experimental

#### Synthesis

We prepared the vertically well-aligned ZnO nanorods grown on a fluorine doped tin oxide (FTO) glass using a hydrothermal method assisted with a ZnO seed layer at 90 °C for 3 hrs, then washed with ethanol, and dried at 100 °C for 1 h.

Surface modification of the ZnO nanorods was conducted by dipping in a fresh 0.1 M TiCl<sub>4</sub> solution mixed with deionized (DI) water under biasing of an ultrasonic power of 750 Watt (20 kHz) at room temperature for several minutes. Surface-modified ZnO nanorods were then washed with DI water, and dried at 70 °C for 1 h. The coating amount was adjusted by changing the dipping time. We fixed the surface modification time at 0, 10, 20, 30 and 40 min. Denomination A, B, C, D, and E implies ZnO nanorod samples prepared under surface modification time of 0, 10, 20, 30 and 40 min, respectively.

#### Fabrication of DSSC

DSSC devices were fabricated with a dye sensitized ZnO nanorods working electrode and a Pt-coated counter electrode, prepared by spin-coating 5 mM  $\rm H_2PtCl_6$  solution onto FTO and sintered at 450 °C for 30 min. Two electrodes were clipped together. They were immersed in 0.5 mmol ethanol solution of N719 dye (Dyesol) at 60 °C for 3 hrs, then washed with ethanol and dried in an oven. Two electrodes were separated by a Surlyn spacer (25  $\mu$ m thick) and sealed by heating. A redox electrolyte solution (Iodolyte AN-50, Solaronix) was introduced between two electrodes.

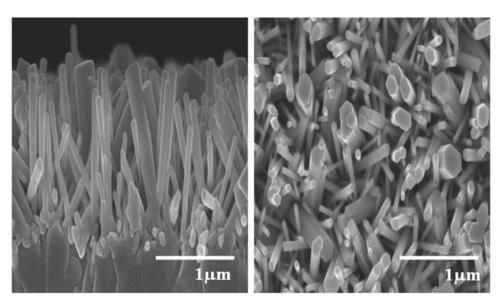
#### Measurements

The comparison and analysis of the material properties between the surface-coated and the non-coated ZnO nanorods was done by the help of FE-SEM (Field Emission Scanning Electron Microscopy, HITACHI S-4800), TEM (Transmission Electron Microscopy, KJ111), XRF (X-ray Fluorescence, Philips PW2400), and UV-VIS-NIR (Ultraviolet-Visible-Near Infrared, Cary 5 spectrophotometer). I-V characteristics of ZnO nanorod-based DSSC device with the active area of  $0.5 \times 0.5$  cm<sup>2</sup> were measured under 100 mWcm<sup>-2</sup> of simulated AM 1.5 solar light by using a Solar Simulator (PEC-L12, Peccell).

#### Results and Discussion

Figure 1 shows the SEM images of the cross section and the top view for ZnO nanorods grown by a hydrothermal process. One can find that the hexagonal nanorods covered the entire surface of the FTO glass. Furthermore, we observed that the ZnO nanorods have the diameter of 50 or 300 nm and the length of about 2  $\mu$ m, showing the preferred orientation in the c-axis direction.

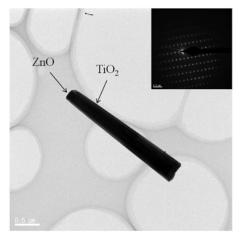
Figure 2 shows TEM image of a TiO<sub>2</sub>-coated ZnO nanorod, choosing a D sample for ease of analysis. The selected-area electron diffraction (SAED) pattern of the ZnO nanorod is shown in inset. We observed that TiO<sub>2</sub> shells of about 20 nm in thickness were coated on the surface of ZnO nanorods. The length and width of ZnO cores are about 2  $\mu$ m and 300 nm, respectively. From the results of the SAED experiments, we observed that the ZnO nanorod has typical single-crystal nature of hexagonal ZnO, which is a crucial factor for photovoltaic device applications.



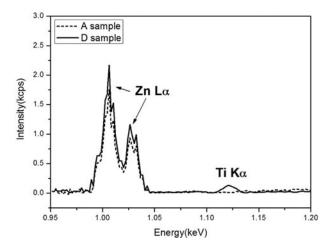
**Figure 1.** SEM images of the cross section and the surface for ZnO nanorods grown by a hydrothermal process.

Figure 3 shows the XRF spectra of the non-coated and TiO<sub>2</sub>-coated ZnO nanorods, corresponding to A sample and D sample, respectively. The XRF spectra reveal that a Ti peak is observed only for the TiO<sub>2</sub>-coated ZnO nanorod, while Zn peaks are observed in both samples. This result confirms in a roundabout way that the nano-sized oxide material coated on the surface of the ZnO nanorod is TiO<sub>2</sub>.

Figure 4 shows the XPS spectra in the Ti 2p and O 1s region for D sample and the peak positions were referenced to carbon. Compared to TiO<sub>2</sub> bulk, the scan results of Ti 2p for the D sample exhibit broader peaks with a binding energy of 458.2 and 463.8 eV



**Figure 2.** TEM image of a TiO<sub>2</sub>-coated ZnO nanorod, choosing a D sample for ease of analysis. The inset figure is SAED pattern of a single ZnO nanorod. The growth direction is parallel to the c-axis.



**Figure 3.** The XRF spectra of ZnO nanorods with and without TiO<sub>2</sub> modification, corresponding to D sample and A sample, respectively.

corresponding to Ti  $2p_{3/2}$  and Ti  $2p_{1/2}$ , respectively, as shown in Fig. 4(a). The O 1s peak of the D sample was also broadened and shifted to a high binding energy, which is likely due to the mixed two peaks, corresponding to Ti-O of TiO<sub>2</sub> or TiOx and OH<sup>-</sup> of Ti(OH)x, as shown in Fig. 4(b). The TiOx formation is due to the low growth temperature, and the Ti(OH)x is formed by reacting TiCl<sub>4</sub> with H<sub>2</sub>O with assisting the sonochemical process. However, our goal is simply to form the nanolayer fast on the ZnO nanorod from TiCl<sub>4</sub> assisted sonochemical process. Therefore, Ti(OH)x and TiOx can be sufficiently applied to DSSC device, like TiO<sub>2</sub>, to increase their photoelectrochemical properties.

Figure 5 shows the UV absorbance spectra of dyes detached from the ZnO nanorods with and without TiO<sub>2</sub> modifications as a function of the surface modification time in TiCl<sub>4</sub> solution under biasing of an ultrasonic power. All spectra exhibited two absorption bands at 376 and 512 nm, indicating characteristics of N719 dye [8]. To estimate the adsorbed amount of dye for five samples, we used the Beer-Lambert law. A rearranged form of the Beer-Lambert law for the ZnO nanorods allows one to calculate the number of moles of

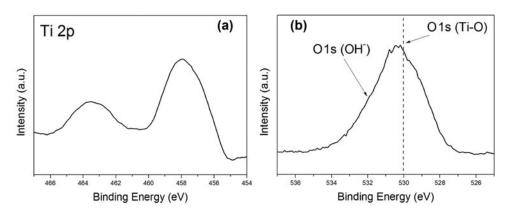


Figure 4. XPS spectra in the (a) Ti 2p and (b) O 1s region for D sample.

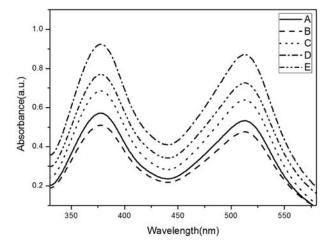


Figure 5. UV absorbance spectra of dyes detached from the ZnO nanorods with and without TiO<sub>2</sub> modifications.

dye adsorbed per 1 cm<sup>2</sup> on the surface of ZnO by the equation,

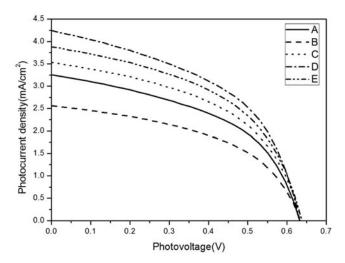
$$\alpha = \varepsilon \cdot 1 \cdot c \tag{1}$$

where  $\alpha$  is the absorbance,  $\varepsilon$  is the molar extinction coefficient of the dye in solution (M<sup>-1</sup>·cm<sup>-1</sup>), l is the path length (cm) and c is the concentration of the sample solution (M). UV-Vis absorption spectra of the resultant solution, whose molar extinction coefficient was determined to be  $1.41 \times 10^4$  dm<sup>3</sup> mol<sup>-1</sup> cm<sup>-1</sup> at 515 nm [9], were measured using a 1 cm quartz cell to estimate the adsorbed amount of dye. Using the given values,  $\varepsilon = 1.41 \times 10^4$  dm<sup>3</sup> mol<sup>-1</sup> cm<sup>-1</sup> at 515 nm, I = 1 cm, and the  $\alpha$  values measured at 515 nm, we can calculate the concentration c by equation (1). The physical characteristics of ZnO nanorods obtained in this way are listed in Table 1. The spectra and calculated concentrations tell that the absorbance increases in the order of B, A, C, E, and D samples due to the surface treatment effects as a function of surface treatment time, implying the thickness effects of TiO<sub>2</sub> shells. However, the B sample has poor absorbance, compared to A sample, which is likely due to the amorphous phase of the TiO<sub>2</sub> shell with thin thickness. Furthermore, the

**Table 1.** Physical characteristics of the ZnO nanorods vs surface modification time

Samples	Surface modification time	Absorbance at 515 nm	Adsorbed N719 <sup>a</sup> (mol·dm <sup>-3</sup> )
A	0 min	0.53	$3.76 \times 10^{-6}$
В	10 min	0.47	$3.33 \times 10^{-6}$
C	20 min	0.64	$4.54 \times 10^{-6}$
D	30 min	0.87	$6.17 \times 10^{-6}$
E	40 min	0.72	$5.11 \times 10^{-6}$

<sup>&</sup>lt;sup>a</sup>Amounts of adsorbed N719 were calculated from the UV-Vis absorption spectra by using Beer-Lambert's law.



**Figure 6.** I-V curves of DSSCs fabricated by using ZnO nanorods with and without TiO<sub>2</sub> modification from TiCl<sub>4</sub> assisted sonochemical process.

absorbance of E sample reduced more than that of D sample, which is probably because of thick thickness of the TiO<sub>2</sub> shell, leading to a reduced surface area.

Figure 6 shows the photocurrent density (I) vs. photovoltage (V) curves of the DSSCs fabricated by using ZnO nanorods with and without TiO<sub>2</sub> modification from a TiCl<sub>4</sub> assisted sonochemical process. The electro-optical characteristics of DSSCs are also listed in Table 2. One can find that there was no significant difference in  $V_{OC}$  between non- and TiO<sub>2</sub>-coated ZnO nanorods-based DSSCs. With increasing TiO<sub>2</sub> surface treatment time of the ZnO nanorods, the energy conversion efficiencies gradually increase, except for E device. A maximum efficiency of 1.37% was achieved in case of D device with short-circuit current density ( $J_{SC}$ ) = 4.25 mA·cm<sup>-2</sup>, open-circuit voltage ( $V_{OC}$ ) = 635.8 mV and fill factor (FF) = 50.7% as listed in Table 2. The enhancement of the energy conversion efficiency originates from the increased  $J_{SC}$  and  $V_{OC}$  in the DSSC devices with ZnO nanorods that are suitably coated TiO<sub>2</sub> shell. These facts can be explained as follows. The TiO<sub>2</sub> nanolayer can form new energy barrier on the ZnO nanorods, which can effectively inhibit the surface charge recombination and improve the energy conversion efficiency of the ZnO nanorod-based dye sensitized solar cells (DSSCs).

Table 2. Electro-optical characteristics of DSSCs with ZnO nanorods

DSSC devices	V <sub>oc</sub> (mV)	J <sub>sc</sub> (mA/cm <sup>2</sup> )	FF (%)	Efficiency (%)
A	635.3	3.30	50.2	1.05
В	634.4	2.57	50.2	0.82
C	635.7	3.55	50.8	1.15
D	635.8	4.25	50.7	1.37
E	635.5	3.87	50.6	1.24

The complex trend in efficiencies can be explained as follows.  $TiO_2$  nanolayer formed by dipping in less than 10min is probably amorphous and remains the dye aggregates in sensitization process, which causes poor electron injection, slow diffusion, and low  $J_{SC}$  values.  $TiO_2$  nanolayers were formed by dipping for 10-30 min, and  $J_{SC}$  value consistently increases due to gradual crystallization of the nanolayers. On the other hand, DSSCs with  $TiO_2$  nanolayer formed by dipping for 40 min reveal a slight decrease in  $J_{SC}$  due to a reduced surface area, leading to smaller dye loading. Therefore, we conclude that DSSCs with  $TiO_2$  nanolayer formed by dipping for 30 min exhibits best efficiencies

#### Conclusions

The ZnO nanorods with the diameter of 50 and 300 nm and the length of 2  $\mu$ m are grown on the FTO glass with a ZnO seed layer by the hydrothermal process, which display the preferred orientation in the c-axis direction. Surface modification on the ZnO nanorods by TiO<sub>2</sub> was carried out by the sonolysis of TiCl<sub>4</sub>, changing modification time. The results demonstrate that an TiO<sub>2</sub> nanolayer can effectively inhibit the surface charge recombination and improve the energy conversion efficiency of the ZnO nanorod-based dye sensitized solar cells (DSSCs). The highest conversion efficiency obtained was 1.37% for the dye-sensitized ZnO nanorods with TiCl<sub>4</sub> modification time of 30 min during the sonochemical process.

#### Acknowledgment

This research was supported by Basic Science Research Program through the National Research Foundation of Korea (NRF) funded by the Ministry of Education, Science and Technology (2013R1A1A4A01008503).

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